Prestressed composite specimen for mixed-mode I/II cracking in laminated materials

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Abstract

This article presents the mixed-mode I/II prestressed end-loaded split specimen for the fracture testing of composite materials. The system is based on the fact that one of the two fracture modes is provided by the prestressed state of the specimen, and the other one is increased up to fracture initiation using a testing machine. The novel beam-like specimen is able to provide any combination of the mode-I and mode-II energy release rates. A simple closed-form solution is developed using beam theory as a data reduction scheme and for the calculation of the energy release rates in the new configuration. The applicability and the limitations of the novel fracture mechanical test are demonstrated using unidirectional glass–polyester composite specimens. If only crack propagation onset is involved, then the prestressed mixed-mode beam specimen can be used to obtain the fracture criterion of transparent composite materials in the $G_{I-}G_{II}$ plane.

Keywords

Delamination, double-cantilever beam, end-loaded split, beam theory

Introduction

The delamination fracture is one of the major damage modes in laminated composite materials. Considering the linear elastic fracture mechanics (LEFM),¹ three basic types of the interlaminar fracture are known: mode-I (opening), mode-II (sliding or in-plane shear), and the mode-III (tearing or anti-plane shear). The significant part of the literature deals with the mode-I, mode-II, and the mixed-mode I/II cases,^{2–4} although the interest of the scientists is subsequently focused on the mode-III and mixed-mode I/II, II/III fractures too.^{5–11}

The currently available mixed-mode I/II configurations were summarized in many papers.^{3,4} Considering the mixed-mode I/II delamination, there are large number of fracture systems, such as the mixed-mode bending (MMB),^{12–14} the asymmetric double-cantilever beam (DCB) specimen,^{15,16} the DCB loaded with uneven bending moments (DCB–UBM),^{17,18} the DCB specimen loaded by forces and moments,¹⁹ and – among others – the Arcan^{20–22} or compact tension shear specimen (CTS).^{23,24} Without any doubt, the most effective one among them is the MMB specimen developed by Reeder and Crews.^{12,13} The specimen is able to provide any mode ratio; however, the complex fixture involved by the MMB specimen is a drawback of the test.

As an alternative possibility, the mixed-mode I/II prestressed end-notched flexure (PENF_{I/II}) system was published in 2006²⁵ and probably it is the simplest configuration, which is ever developed. The system requires only a three-point bending setup and few steel rollers with different diameters. The rollers are used to prestress the specimens by fixing the crack opening displacement (COD); then we can put the specimen into a three-point bending setup and the mode-II energy release rate (ERR) can be increased up to the

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critical value. In Szekrényes²⁵, the applicability and the drawbacks of the test are demonstrated using glass–polyester composite specimens. It was shown that the mode ratio changes with the crack length and the external load, although this change is estimated to be small. An improved beam theory (IBT) scheme was found to be accurate for the data reduction and the fracture envelope of the material was obtained using the well-known power and Williams' criteria.

The method of prestressed composite beams was later extended to mixed-mode II/III^{26} and I/III cases²⁷ too.

The aim of this article is to develop a similar mixed/ mode I/II configuration, which makes it possible to test the material at relatively large crack lengths. This is the reason for why the combination of the end-loaded split (ELS),^{28,29} and the DCB,^{30,31} is realized. While in the case of the $PENF_{I/II}$ the crack length range is limited by the central load introductor, in the ELS specimen the end load does not make any restrictions to the crack length. The PELS_{I/II} is based on the same concept as the $PENF_{I/II}$. A steel roller is inserted between the specimen arms and the built-in prestressed specimen is loaded by a force at the end. The applicability and the advantages/drawbacks of the test are demonstrated by finite element (FE) calculations and experiments. Finally, a fracture envelope is obtained in the G_{I} - G_{II} plane.

The PELS specimen for mixed-mode I/II cracking

The PELS_{I/II} specimen is the combination of the DCB and ELS specimens. As it is shown in Figure 1, P_{DCB} is the load related to the mode-I part of the ERR caused by a steel roller (Figure 1(c)), while P_{ELS} is the load related to the mode-II loading. For the analysis of the PELS_{I/II} configuration, we superimpose the solutions of the DCB and ELS specimens. In a previous work, the IBT solutions for the DCB and ELS specimens were presented.³²

The compliance of the DCB specimen can be expressed as: 32

$$C_{\rm DCB} = \frac{8a^3}{bh^3 E_{11}} + \frac{2a^3}{bh^3 E_{11}} \left(f_{W1} + f_T + \frac{f_{\rm SV}}{2} \right) \tag{1}$$

where *a* is the crack length, *b* the specimen width, *h* the half thickness, and E_{11} the flexural modulus of the specimen. Furthermore, f_{W1} is obtained from the Winkler–Pasternak foundation, f_T from transverse shear and f_{SV} accounts for the Saint–Venant effect:

$$f_{W1} = 5.07 \left(\frac{h}{a}\right) \left(\frac{E_{11}}{E_{22}}\right)^{\frac{1}{4}} + 8.58 \left(\frac{h}{a}\right)^{2} \left(\frac{E_{11}}{E_{22}}\right)^{\frac{1}{2}} + 2.08 \left(\frac{h}{a}\right)^{3} \left(\frac{E_{11}}{E_{22}}\right)^{\frac{3}{4}},$$
 (2)

$$f_T = \frac{1}{k} \left(\frac{h}{a}\right)^2 \left(\frac{E_{11}}{G_{12}}\right),\tag{3}$$

$$f_{\rm SV} = \frac{12}{\pi} \left(\frac{h}{a} \right) \left(\frac{E_{11}}{G_{12}} \right)^{\frac{1}{2}},\tag{4}$$

where E_{22} is the transverse modulus in the y direction, G_{12} the shear modulus of the material in the x-y plane and k = 5/6 the shear correction factor. The mode-I ERR of the DCB specimen can be obtained using the Irwin–Kies expression:¹

$$G_{\rm C} = \frac{P^2}{2b} \frac{\mathrm{d}C}{\mathrm{d}a},\tag{5}$$

that is:

$$G_I = \frac{P_{\rm DCB}^2 a^2 (12 + f_{W2} + f_T + f_{SV})}{b^2 h^3 E_{11}},\tag{6}$$

where P_{DCB} is the external load, and:

$$f_{W2} = 10.14 \left(\frac{h}{a}\right) \left(\frac{E_{11}}{E_{22}}\right)^{\frac{1}{4}} + 8.58 \left(\frac{h}{a}\right)^2 \left(\frac{E_{11}}{E_{22}}\right)^{\frac{1}{2}}.$$
 (7)



Figure 1. The mixed-mode I/II PELS specimen (c) as the superposition of the DCB (mode-I) (a) and ELS (mode-II) (b) specimens.

The COD (δ_{DCB}) of the specimen can be controlled by inserting a steel roller between the specimen arms. A reasonable assumption is that the COD is approximately equal to d_0 . For this reason, we express the force which arises in the DCB specimen using Equation (1) and the definition of $C_{DCB} = \delta_{DCB}/P_{DCB}$:

$$P_{\rm DCB} = \frac{bh^3 E_{11} \delta_{\rm DCB}}{8a^3} \frac{1}{1 + (f_{W1} + f_T + f_{SV}/2)/4}, \quad (8)$$

where $\delta_{\text{DCB}} = d_0$ is the diameter of the prestressing roller. Substituting Equation (8) into Equation (6) we obtain:

$$G_{\rm I}^{\rm DCB} = \frac{h^3 E_{11} \delta_{\rm DCB}^2}{64a^4} \frac{[12 + f_{W2} + f_T + f_{SV}]}{[1 + (f_{W1} + f_T + f_{SV}/2)/4]^2}.$$
 (9)

The improved solution for the ELS specimen's compliance is:³²

$$C^{\text{ELS}} = \frac{3a^3 + L^3}{2bh^3 E_{11}} + \frac{L}{2bhkG_{13}} + \frac{3}{\pi} \frac{L^2}{2bh^2 E_{11}} \left(\frac{E_{11}}{G_{13}}\right)^{\frac{1}{2}} + \frac{a^3}{2bh^3 E_{11}} \left[0.98 \left(\frac{h}{a}\right) \left(\frac{E_{11}}{G_{13}}\right)^{\frac{1}{2}} + 0.43 \left(\frac{h}{a}\right)^2 \left(\frac{E_{11}}{G_{13}}\right) \right],$$
(10)

and the ERR becomes:

$$G_{\rm II}^{\rm ELS} = \frac{P^2 a^2}{4b^2 h^3 E_{11}} [9 + f_{\rm SH2}],\tag{11}$$

where f_{SH2} is obtained from the crack tip deformation:³²

$$f_{\rm SH2} = 1.96 \left(\frac{h}{a}\right) \left(\frac{E_{11}}{G_{13}}\right)^{\frac{1}{2}} + 0.43 \left(\frac{h}{a}\right)^2 \left(\frac{E_{11}}{G_{13}}\right)^{\frac{1}{2}}$$

Finally, the mode ratio of the PELS_{I/II} specimen is:

$$\frac{G_{\rm I}}{G_{\rm II}} = \frac{b^2 E_{11}^2}{16} \left(\frac{h}{a}\right)^6 \left(\frac{\delta_{\rm DCB}}{P_{\rm ENF}}\right)^2 \\
\times \frac{(12 + f_{W2} + f_T + f_{SV})}{\left[1 + (f_{w1} + f_T + f_{SV}/2)/4\right]^2} \frac{1}{(9 + f_{SH2})}, \quad (12)$$

which depends on the crack length and the applied load also. All of the factors in Equation (12) have been defined before. The accuracy of the analytical solution has already been proven in previous papers.^{25,32} However, some other questions should be clarified and these are detailed in the following sections.

The dependence of the mode ratio on the crack length can be investigated based on Equation (12). In Table 1, we calculated the mode ratio at a = 105 mm

Table I. The changes in the mode ratio G_I/G_{II} in the case of inaccurate crack initiation detection if a = 105 mm

Δa	0.25	0.35	0.5	0.75	I	1.25	١.5	2
Difference (%) ^a	1.4	1.9	2.7	4.1	5.4	6.7	8.0	10.4

 $^{a}(G_{I}/G_{II}|_{a+\Delta a}-G_{I}/G_{II}|_{a=105})/(G_{I}/G_{II}|_{a=105}).$

using the following material properties: $E_{11} = 33$ GPa, $E_{22} = 7.2$ GPa, and $G_{12} = 3$ GPa. The geometrical properties are: b = 20 mm and 2h = 6.2 mm. In the calculation, we subsequently increased the crack length by small increments until 2 mm. If the crack length is measured inaccurately, it has a moderate influence on the mode ratio. If $\Delta a = 1$ mm, then the expected error is 5.4%. This indicates that the accurate measurement of the crack length after crack initiation is important. During the experiments (in a later section), a transparent material is used, so that the crack initiation is easily identified within 0.5 mm.

If the crack opening is relatively large, then the crack lengths of the upper and lower specimen arms will be different from their original values. This is due to the fact that the contact point between the specimen arms and the roller is located at a distance of a_{11}^* and a_{11}^* instead of 'a' from the crack tip, as it is shown in Figure 2. We can calculate the distances, a_{μ}^* and a_{μ}^* . based on a simple geometrical analysis. According to Figure 2, we need to find the coordinates, $a_{\rm u}^*$ and $a_{\rm l}^*$, where the derivative of $f_i(x) = \pm (x + 2a)(x - a)^2/a$ $(4a^3) \cdot \delta_{\text{DCB}} + \frac{1}{2}(4x^3 - 9a^2x - 3xL + 6a^3 + 2L^3)}{(3a^3 + 3a^2x - 3xL + 6a^3 + 2L^3)}$ L^{3}) δ_{ELS} (simple beam theory solution, i=1 for the upper arm, the positive sign, i=2 for the lower arm, negative sign) and $f_{c}(x) = ((\delta_{\text{DCB}}/2)^{2} - x^{2})^{1/2} + \delta_{\text{FLS}}$ (equation of a circle with diameter, δ_{DCB}) is the same. Table 2 shows the changes of the crack length, and the difference compared to the original crack length. In the worst case (if $d_0 = 13$ mm), we make a mistake of 1% in the crack length. A reasonable conclusion is that the change of the crack length is negligible.

A major question in a mixed-mode configuration is the estimation of the change in the mode ratio along the crack front, and its effect on the ERRs. To clarify this question, a FE model shown in Figure 3 is utilized. For the calculation, the ANSYS 12 package was used. The elastic properties of the models were: $E_{11} = 33$ GPa, $E_{22} = E_{33} = 7.2 \text{ GPa}, \quad G_{12} = G_{13} = G_{23} = 3 \text{ GPa},$ and $v_{12} = v_{13} = v_{23} = 0.27$. The geometrical properties are: b = 20.0 mm, 2h = 6.2 mm and the length of the models was L = 150 mm (Figure 1). The 3D model of the ELS specimen was built using linear eight-node solid brick elements. The imposed boundary conditions and the loading of the model are demonstrated in Figure 3. At the end of the specimen arms, the displacements shown in Figure 3 were imposed, where δ_{DCB} was equal to the actual roller diameter ($d_0 = 6, 7, 8, 10, 12$,



Figure 2. The effect of the prestressing roller on the crack length of the PELS_{1/II} specimen.

Table 2. The changes in the crack length of the prestressed specimen with the steel roller diameter

δ _{DCB} (mm)	0	6	7	8	10	12	13
δ_{ELS} (mm)	19.4	16.1	12.8	11.0	8.8	5.6	4.2
<i>a</i> [*] _u (mm)	105	106	105	105	105	105	105
Difference compared to $a = 105 \text{ mm}$ (%)	0.0	1.0	0.0	0.0	0.0	0.0	0.0
<i>a</i> [*] _l (mm)	105.0	104	104	104	104	104	104
Difference compared to $a = 105 \text{ mm}$ (%)	0.0	-1.0	-1.0	-1.0	-1.0	-1.0	— I .C

 a_{u}^{*} , the crack length of the upper arm; a_{l}^{*} , the crack length of the lower arm.



Figure 3. The applied boundary conditions of the $PELS_{I/II}$ specimen and the details of the FE model.

size of the crack tip elements were $\Delta x = \Delta y = 0.25 \text{ mm}$ and $\Delta z = 1 \text{ mm}$ (Figure 3, coordinate system).

In the crack tip, a refined mesh was constructed and the mode-I and mode-II ERRs were evaluated by using the virtual crack-closure technique (VCCT),^{33,34} the

Figures 4(a) and (b) and 5(a) show the distribution of the ERRs along the crack front in the case of $d_0 = 8$, 10, and 12 mm, while Figure 5(b) shows the variation of



Figure 4. The distribution of the mode-I and mode-II ERRs along the crack front in the case of $d_0=8.0$ mm (a) and $d_0=10.0$ mm (b).



Figure 5. The distribution of the mode-I and mode-II ERRs along the crack front in the case of $d_0 = 12.0$ mm (a) and the variation of the mode ratio along the crack front if $d_0 = 8.0$, 10.0, and 12.0 mm (b).

the mode ratio in each case. The values of δ_{ELS} were determined based on crack initiation experiments (in a later section). As it is expected, the mode-I and mode-II ERRs have a symmetric distribution along the crack front (similarly to the results in³⁵). In Figures 4(a) and (b) and 5(a) and (b), the average ERRs and mode ratios were obtained by dividing the integrated area under the curve by the specimen width. Table 3 shows the comparison of the IBT to the VCCT with respect to the average ERR and mode ratio. The IBT overestimates G_{I} at most with 4.3% and even G_{II} at most with 11.9%. This relatively significant difference

is due to the fact that the FE model was loaded by imposed displacement, and the δ_{ELS} displacement was calculated based on the measured load-displacement slopes and the critical P_{ELS} values. The maximum difference in the mode ratio is 12.9%, which is the result of the rounding of G_{I} and G_{II} . Overall, this agreement is good and based on these results, the IBT is a possible data reduction scheme for the PELS_{I/II} test.

Since the ERR varies along the crack front, that is, it is the highest at the center and the lowest at the specimen edges, it is obvious that the specimen possesses a curved crack front under crack propagation. As it is

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Roller diameter	<i>d</i> ₀ (mm)	0 (ELS)	6	7	8	10	12	13	\sim I5 (DCB)
Load at crack initiation	P _{ELS} (N)	103.1	82.6	70.8	61.9	48.3	31.3	22.5	0
Displacement at crack initiation	δ_{ELS} (mm)	19.4	16.1	12.8	11.0	8.8	5.6	4.2	0
G _I (J/m ²)	IBTª	_	48	65	85	132	190	223	412
	VCCT widthwise average ^b	-	47	63	83	129	186	218	395
	VCCT at z^c_{max}	_	54	73	96	149	215	253	457
	Difference (a–b)/b (%)	_	2.1	3.2	2.4	2.3	2.2	2.3	4.3
	Difference (<i>a–c</i>)/c (%)	-	-11.1	-11.0	-11.5	-11.4	-11.6	-II.9	-9.8
G _{II} (J/m ²)	IBT ^a	707	454	333	255	156	66	34	-
	VCCT widthwise average ^b	724	502	316	234	151	59	34	-
	VCCT at z_{max}^{c}	767	533	336	249	160	63	36	-
	Difference (a–b)/b (%)	-2.3	-9.6	5.4	9.0	3.3	11.9	0.0	-
	Difference (<i>a–c</i>)/c (%)	-7.8	-14.8	-0.9	2.4	-2.5	4.8	-5.6	-
G _I /G _{II}	IBTª	0	0.11	0.20	0.33	0.85	2.88	6.56	∞
	VCCT widthwise average ^b	0	0.09	0.20	0.35	0.85	3.15	6.41	∞
	VCCT at z_{max}^{c}	0	0.10	0.22	0.39	0.93	3.41	7.03	∞
	Difference (<i>a</i> – <i>b</i>)/ <i>b</i> (%)	-	12.9	-2.I	-6.I	-1.0	-8.7	2.3	-
	Difference (<i>a</i> – <i>c</i>)/ <i>c</i> (%)	_	4.4	-10.2	-13.5	-9.I	-15.6	-6.7	_

Table 3. Comparison of the ERRs and mode ratios by beam and FE analysis

^aIBT; ^bVCCT widthwise average; and ^cVCCT at z_{max}.

seen in Figure 5(b), it is not possible to produce a constant mode ratio along the crack front. Consequently, some assumptions are required considering the reduction of the experimental data. In the data reduction and the calculation of G and $G_{\rm I}/G_{\rm II}$, the widthwise average values will be adopted. To assess the possible errors (Figures 4(a) and (b) and 5(a)) in the calculation of Gand $G_{\rm I}/G_{\rm II}$ by the IBT we assume that the crack initiation takes place at the point where the total ERR $(G_{\rm T})$ reaches the highest value. This point can be found always in the midpoint of the crack fronts marked by z_{max} . In Table 3, the ERRs and mode ratios are calculated by using the values by VCCT at z_{max} too and they are compared to the result of IBT. The difference in G_{I} is around 11.5%, while in the case of $G_{\rm II}$ the difference changes between 1% and 15%, which seems to be significant. On the contrary, the difference in the mode ratio is negligible in each case.

Although there are some errors in the analytical predictions of $G_{\rm I}$ and $G_{\rm II}$, the mode ratio is accurately estimated by the IBT method, and we prefer the IBT as a data reduction scheme, because the FE analysis-based data reduction requires much computational time. Moreover, assumption is also required if we apply the VCCT to reduce the experimental data, and the IBT agrees excellently with the widthwise average *G* and $G_{\rm I}/G_{\rm II}$. It should be mentioned that a similar variation of the mode ratio could exist in the MMB system too.¹²

In the sequel, the details of the experimental work is presented.

Experiments

Material properties

The constituent materials of the E-glass–polyester composite were procured from Novia Ltd. The properties of the E-glass fiber are E = 70 GPa and v = 0.27, and the properties of the unsaturated polyester resin we applied are: E = 3.5 GPa and v = 0.35. Both were considered to be isotropic. The unidirectional ($[0^{\circ}]_{14}$) E-glass–polyester specimens with thickness of 2h = 6.2 mm, width of b = 20 mm, and fiber–volume fraction of $V_f = 43\%$ were manufactured in a special pressure tool. A polyamide (PA) insert with thickness of 0.03 mm was placed at the midplane of the specimens to make an artificial starting defect. It should be highlighted that this film thickness is higher than the 0.013 mm recommended in the DCB and MMB test standards.^{2,3}

A great advantage of the E-glass-polyester material used in this study is its transparency, which makes it possible to observe the crack initiation visually. The tool was left at room temperature until the specimens became dry. Then the specimens were removed from the tool and were further left at room temperature until 4–6 h. The specimens were cut to the desired length and were precracked in opening mode of 4–5 mm by using a sharp blade.

The flexural modulus of the material was determined from a three-point bending test with span length of 2L = 150 mm using six uncracked specimens with

thickness of 2h = 6.2 mm and width b = 20 mm. The flexural modulus was computed in accordance with simple beam theory expression: $\delta_{\text{bend}} = PL^3/(4bh^3E_{11})$, which is the displacement at the point of load introduction. The experiments resulted in $E_{11} = 33$ GPa. In fact, the shear deformation was not accounted for; however, its contribution to the displacement is quite small: $\delta_{\text{shear}} = PL/(4bhkG_{12})$. The additional properties were predicted from simple rules of mixture; in this way $E_{22} = E_{33} = 7.2 \text{ GPa}, G_{12} = G_{13} = 3 \text{ GPa}, \text{ and } v_{12} = v_{13} = 0$ 0.27 were obtained. Using these values, the ratio of the displacements from bending and shear deformation at the center of the specimens is: $(h/L)^2 \cdot E_{11}/(G_{12}k) =$ 0.0056, that is, the amount of shear deformation is quite small and can be neglected. The material properties were used in the data reduction process.

DCB test

In the case of the DCB test (Figure 1(b)), we refer to previous fracture experiments³⁶ performed in the crack length range of a = 20-150 mm. It has been found that the critical ERR at crack initiation becomes independent of *a*, after 90 mm. The steady-state ERR was $G_{\rm IC} = 412 \text{ J/m}^2$ evaluated by using an IBT scheme. This value will be used in the sequel.

ELS test

For the ELS measurements, four specimens were prepared with a = 105 mm. Each specimen was put into a clamping fixture and then the specimens were loaded, and the load and displacement values were read from the scale of the testing machine and using a digitronic indicator, respectively. The crack initiation was identified visually and when the first non-uniformity in the previously straight crack front was observed, it was believed to be the point of crack initiation.

PELS test

The PELS_{I/II} tests were carried out using an Amsler testing machine under displacement control. The crack length of interest was a = 105 mm. The critical COD measured from the DCB test³⁶ is about 15 mm (if a = 105 mm). Six steel rollers were used including the following diameters: $d_0 = 6$, 7, 8, 10, 12, and 13 mm. It was assumed that the CODs (δ_{DCB}) were identical to these values. The specimen arms transmitted a relatively high pressure to the steel roller; therefore, the position of the rollers was always stable and no slip along the x axis was observed during the measurements. Similar to the ELS tests, we applied four coupons at each steel roller. The load–deflection data was measured by using the scale of the testing machine and a digitronic indicator. In each case, the critical load at crack initiation was determined.

Results and discussion

It will be shown subsequently that the stiffness, the compliance, and the mode-II ERR of the $PELS_{I/II}$ specimen are identical to those of the ELS specimen.

Load and displacement

Figure 6(a) shows a recorded load-displacement curve for the PELS_{I/II} specimen in the case of $d_0 = 10.0$ mm. The response follows essentially a linear relation. The PELS_{I/II} test was performed according to the following procedures. The onset of crack advance was identified by visual observations. In each case, four specimens were tested, and one of them was used to investigate the crack front. The other three specimens were loaded continuously and the crack initiation was observed in situ. Thus, the former specimen was loaded subsequently, at some points where the initiation was expected the specimen was relieved, and removed from the rig, and the crack front was photographed. When the first non-uniformity was observed, this point was denoted as the point of fracture initiation. The results of this process are demonstrated in Figure 6(b) for the PELS_{I/II}, when the diameter of the prestressing roller was $d_0 = 10.0$ mm.

Table 4 shows that the slopes of the load–displacement traces of the ELS ($d_0 = 0$) and PELS_{I/II} specimens are eventually the same. Consequently, the prestressed state does not influence the stiffness of the system noticeably, and the compliance of the PELS_{I/II} can be assumed to be equal to that of the ELS system. The maximum difference is 8.6% between the measured and calculated slopes. On the other hand, the results in Table 5 confirm that the effect of friction between the steel roller and the specimen arms on the load–displacement response of the system is insignificant.

Data reduction

Two reduction techniques (IBT and direct beam theory (DBT)) were applied to reduce the experimental data. In a relevant work²⁵ for the mixed-mode I/II version of the PENF specimen, four reduction schemes were utilized: IBT, simple beam theory, DBT,³⁷ and the compliance calibration (CC). For mode-I and mode-II specimens, the CC is thought to be the most accurate one; however, its application requires much experimental work. That is why the IBT is the optimal solution in unidirectional composites.

Improved beam theory. Double-cantilever beam: In Equation (6), P_{DCB} should be replaced with P_{I}



Figure 6. Load-displacement curve of the PSCB_{1/111} system if $d_0 = 10 \text{ mm}$ (a). The identification of crack initiation during the fracture process (b).

(the load value at crack initiation in the DCB specimen) in order to obtain the improved analytical expression for the ERR of the DCB specimen ($G_{IC} = 412 \text{ J/m}^2$ from IBT³⁶).

ELS specimen: Replacing P_{ELS} with P_{II} in Equation (11) gives the improved solution for the ELS coupon, where P_{II} is the critical load value at crack onset. The IBT resulted in $G_{\text{IIC}} = 707 \pm 33 \text{ J/m}^2$.

PELS specimen: The improved analytical solutions are given by Equations (9) and (11) for the $PELS_{I/II}$ system.

Direct beam theory. Double-cantilever beam: For the DCB specimen, we replace the result of DBT with the one by IBT ($G_{IC} = 412 \text{ J/m}^2$).

ELS specimen: In accordance with DBT, it is possible to obtain the following scheme for the ELS specimen:³⁷

$$G_{\rm DBT} = \frac{9P_{\rm II}\delta_{\rm ELS}a^2}{2b(3a^3 + L^3)}.$$
 (13)

where P_{II} and δ_{ELS} are the experimentally measured load and displacement values at the point of crack initiation in the ELS specimen.

PELS specimen: We obtain the DBT scheme for $G_{\rm II}$, the PELS_{I/II} system if we replace $P_{\rm II}$ with $P_{\rm ELS}$ in Equation (13) ($G_{\rm I}$ is calculated by IBT). A significant advantage of the DBT is that there is no need to determine the elastic properties of the material.

d ₀ (mm)	0	6	7	8	10	12	13
Load–displacement curve slope-1 (N/mm)	5.5	5.4	5.6	5.5	5.2	5.3	5.7
Difference compared to 5.377 N/mm ^a	1.6	0.6	3.6	1.6	-3.6	-1.9	5.2
Load–displacement curve slope-2 (N/mm)	5.0	5.1	5.4	5.7	5.4	5.5	5.8
Difference compared to 5.377 N/mm ^a	-7.7	-5.5	0.5	5.3	1.1	2.2	8.6

Table 4. The changes in the slope of the load-displacement curves of the PELS_{I/II} specimen with the roller diameter

^aResult of the analytical model.

Table 5. Critical ERRs in the PELS_{I/II} specimen calculated by DBT

	d ₀ (mm)	0 (ELS)	6	7	8	10	12	13	\sim I5 (DCB)
	G _I /G _{II}	0.00	0.10	0.21	0.37	0.91	3.17	6.91	∞
		_	±0.01	±0.02	±0.03	±0.16	±0.93	±0.86	_
DBT	G _I (J/m ²)	0.0	51	70	92	143	206	242	412 (IBT)
	G _{II} (J/m ²)	735	490	334	251	157	65	35	0.0
		±52	±36	±28	±23	±24	±19	±4	_
	GT	735	541	404	342	300	271	277	412 (IBT)

Critical ERRs

The critical mode-I, mode-II, and the mixed-mode I/II ERRs at crack initiation and the mode mix calculated by the IBT are given in Table 3. The geometries tested had properties of $a = 105 \,\mathrm{mm}, 2h = 6.2 \,\mathrm{mm},$ and L = 150 mm, and at each value of the diameter of the prestressing roller (d_0) four coupons were used. Table 5 presents the results obtained by the DBT scheme. In fact, the scatter of the mode-I ERR component is zero; this is because the mode-I ERR is provided by the preload of the specimen. Otherwise, the mode-I components were substituted by the results of the IBT scheme. Comparing Tables 3 and 5, it is elaborated that the difference between the G_{II} values – as well as the mode ratio - by DBT and IBT decreases with the roller diameter (which is eventually equal to the COD). Furthermore, the DBT method gives G_{I} and G_{II} values which are closer to the maximum values calculated by the FE model. The mode ratio is predicted by both methods in an equally accurate manner.

It is important to recommend a data reduction technique for the $PELS_{I/II}$ system. The reliability and simplicity of the IBT has already been highlighted by other authors.^{37,38} On the other hand, the application of the FEM as a data reduction method requires large computational time. Although the DBT method agrees quite well with the VCCT method, at the midpoint of the crack front the IBT seems to be more conservative method in each mode ratio. At this stage, the optimal solution is the application of IBT for the evaluation of both the mode-I and mode-II ERRs; however, the theoretically possible errors should be considered (section 2 and Table 3).

Fracture envelopes

In order to construct a fracture envelope in the $G_{\rm I}$ – $G_{\rm II}$ plane, we apply two criteria. In accordance with the traditional power criterion, the following relation may be established between the mode-I and mode-II ERRs:^{39, 40}

$$\left(\frac{G_{\rm I}}{G_{\rm IC}}\right)^{p_1} + \left(\frac{G_{\rm II}}{G_{\rm IIC}}\right)^{p_2} = 1.$$
(14)

Williams' criterion³⁹ recommends the following expression for the establishing relation between the mode-I and mode-II ERRs:

$$\left(\frac{G_{\rm I}}{G_{\rm IC}} - 1\right) \left(\frac{G_{\rm II}}{G_{\rm IIC}} - 1\right) - I_i \left(\frac{G_{\rm I}}{G_{\rm IC}}\right) \left(\frac{G_{\rm II}}{G_{\rm IIC}}\right) = 0, \quad (15)$$

where I_i is the interaction parameter between the mode-I and mode-II ERRs. If $I_i = 0$, then there is no interaction. Also, if $I_i = 1$, then Equation (15) states a simple addition. In Equations (14) and (15), G_{IC} is the critical ERR under pure mode-I (calculated from the data of the DCB specimen) and G_{IIC} is the mode-II critical ERR (calculated from the data of the ELS specimen). The results of the PELS_{I/II} test listed in Table 3 (IBT) were used to provide six additional points in the G_{I} - G_{II} plane. The power parameters (p_1, p_2) in Equation (14) and the interaction parameter (I_i) in Equation (15) were determined by a curve-fit technique by applying the OriginPro 7.0 code.



Figure 7. Interlaminar fracture envelope in the $G_{\Box}-G_{\Box}$ plane for E-glass–polyester composite material calculated by the IBT method.



Figure 8. Interlaminar fracture envelope in the G_I-G_{III} plane for E-glass-polyester composite material calculated by the IBT method.

The fracture envelope calculated by the IBT method is displayed in Figure 7. The main conclusion is that there is some interaction between the mode-I and mode-II ERRs. The scatter is also in reasonable ranges. Overall, the difference between the power and Williams' criteria is negligible; both describe the same failure locus.

In a recent work, the fracture envelopes in the G_{I} - G_{III} plane were constructed by the mixed-mode I/III prestressed split-cantilever beam (PSCB)_{I/III} specimen for the same E-glass–polyester material.²⁷ A similar experimental study resulted in a convex envelope in the $G_{\rm I}$ – $G_{\rm III}$ plane as it is shown in Figure 8. Based on the comparison between Figures 7 and 8 we may conclude that the material proves similar behavior in the $G_{\rm I}$ – $G_{\rm II}$, and the $G_{\rm I}$ – $G_{\rm III}$ plane, since interaction takes place in both cases.

Conclusions

In this study, the mixed-mode I/II version of the PELS specimen was developed for interlaminar fracture testing of laminated transparent composite materials. Apart from the ELS and the traditional DCB tests, the PELS_{I/II} specimen was used to obtain the mixed-mode I/II ERR at crack propagation onset including six different mode ratios. To perform the experiments, unidirectional E-glass–polyester specimens were manufactured. An improved beam model was recommended for the evaluation of both the mode-I and mode-II ERRs.

A FE analysis was performed and it was shown that the mode ratio changes along the specimen width and it is not possible to eliminate this variation. The beam theory expressions give approximately the widthwise average value of the ERRs and mode ratio compared to the FE results. In this respect, some assumptions were made considering the data reduction, namely the widthwise average values were believed to give acceptable and realistic results. To assess the possible errors committed through these assumptions, a FE analysis was performed using the VCCT. The crack initiation was expected at the point where the maximum of the total ERR was calculated and the values of G_{II} and G_{I}/G_{II} were compared to the widthwise average values. The possible errors were estimated to be at most 11.9% in G_{II} and 15.6% in G_{I}/G_{II} ; however, these differences are the results of the rounding.

Based on the performed experimental study, the fracture envelope of the present material was determined indicating a significant interaction between G_{II} and G_{II} .

The PELS_{I/II} specimen offers several advantages. First, it incorporates the traditional beam-like specimen geometry. Second, it was shown that the PSCB_{I/II} specimen is able to produce any mode ratio at crack propagation onset. The drawbacks of the PELS_{I/II} specimen are that the mode ratio changes with the crack length and the applied load; so the method is recommended mainly for the testing of transparent composite materials. Moreover, the mode ratio changes even along the crack front. Finally, the mode ratio cannot be calculated without performing experiments (i.e., it cannot be designated before the test), involving the fact that the mode ratio will depend on the definition of the crack initiation and the accuracy of the measurement of the load and crack length.

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